Calcium phosphate based biomaterials: solid state characterization

Gigliola Lusvardi







NIS Colloquium Advances in biomaterials: combining simulations and experiments

Torino 28-29 November 2013



Mineral phase is composed mainly of microscopic crystals of calcium phosphates, in which the hydroxyapatite (HA), whose chemical formula is Ca10(PO4)6(OH)2, is the most important.

Other mineral phases are dicalcium phosphate $(Ca_2P_2O_7)$, dibasic calcium phosphate (DCP, CaHPO₄), tricalcium phosphate (TCP, Ca₃(PO₄)₂) and some amorphous phases of calcium phosphate.

Biological apatites

 $(Ca, Na, Z)_{10}(PO_4, CO_3, Y)_6(OH, X)_2$

Z = Sr²⁺,Ba²⁺,Pb²⁺, K⁺ Y= HPO₄²⁻ X= F⁻, Cl⁻ Of all the calcium phosphate ceramics used as bone-replacement materials, HA most closely resembles the main inorganic phase of bone and teeth in humans.

HA: very insoluble compound and, under physiological conditions of temperature and pH, is the most insoluble calcium phosphate

Hydroxyapatite (HA) , Ca₁₀(PO₄)₆(OH)₂

hexagonal system, although with some exception in a monoclinic system

 $P6_{3/}m$

a=b=9.418 Å , c=6.884 Å

•The structure can easly accomodate a great variety of substitutions, both cationic and anionic

•The incorporation of foreign ions affect the crystallinity, morphology, lattice parameters and as a consequence the stability of the structure





Confocal micro-Raman spectroscopy

•Fourier transform infrared (FT-IR) adsorption spectroscopy; attenuated total reflectance (ATR) spectroscopy

Scanning electron microscopy-energy dispersive spectroscopy (SEM-EDS)

•Specific surface area (SSA) measurements

•Thermal analysis (DSC/DTA)

Transmission electron microscopy (TEM)

•X-ray photoelectron spectroscopy (XPS)

·X-ray powder diffraction (XRPD)



Contents lists available at SciVerse ScienceDirect

MATERIAL

Materials Science and Engineering C

journal homepage: www.elsevier.com/locate/msec

Sr-containing hydroxyapatite: morphologies of HA crystals and bioactivity on osteoblast cells

Valentina Aina ^{a,b,c,1}, Loredana Bergandi ^{d,*,1}, Gigliola Lusvardi ^e, Gianluca Malavasi ^e, Flora E. Imrie ^f, Iain R. Gibson ^f, Giuseppina Cerrato ^{a,b,c}, Dario Ghigo ^d

^a Department of Chemistry, Università degli Studi di Torino, Via P. Giuria 7, 10125 Torino, Italy

^b Centre of Excellence NIS (Nanostructured Interfaces and Surface) Università degli Studi di Torino

^c INSTM (Italian National Consortium for Materials Science and Technology), UdR Università di Torino

^d Department of Oncology, Università degli Studi di Torino, Via Santena 5/bis, 10126 Torino, Italy

^e Department of Chemical and Geological Sciences, Università di Modena and Reggio Emilia, Via Campi 183, 41125 Modena, Italy

¹ School of Medical Sciences, Institute of Medical Sciences, University of Aberdeen, Foresterhill, Aberdeen, AB25 2ZD, United Kingdom

Strontium : improved solubility, antiresorptive activity, osteoclast apoptosis and osteoblast stimulation.

It shows that a great number of papers on the synthesis (with different methods) and physico-chemical characterization of Sr-HA have been reported, but with *often discordant data*

Sr-HA obtained by a solid state method

Number of moles of reactants used and metal/phosphorus molar ratios in the solidstate synthesis of strontium substituted hydroxyapatite.

Desired composition	No. of mole	es of reactan	M/P molar ratio	
	CaHPO ₄	CaCO ₃	SrCO ₃	(M = Ca + Sr)
Ca10(PO4)6(OH)2	0.06	0.04	-	1.67
$Ca_8Sr_2(PO_4)_6(OH)_2$	0.06	0.02	0.02	1.67
$Ca_6Sr_4(PO_4)_6(OH)_2$	0.06	-	0.04	1.67



1) $Ca_6Sr_4(PO_4)_6(OH)_2$ 2) $Ca_8Sr_2(PO_4)_6(OH)_2$ 3) $Ca_{10}(PO_4)_6(OH)_2$

Sr-HA was formed for x=2

Sr-HA with a small amount of β -TCP for x=4

$Ca_{(10-x)}Sr_{x}(PO_{4})_{6}(OH)_{2}$

Nominal composition	(hkl)	Intensity [counts]	d [Å]	Position [°20]	FWHM [°20]	i = b[Å]	(Å]	Crystallite size d [Å]	Crystallinity degree (X_c) [%±4]
Ca10(PO4)6(OH)2	(211)	25611	2.81	31.84	0.10	9.413(3)	6.879(5)	817 ± 25	93
	(0 0 2)	8888	3.44	25.93					
	(300)	14533	2.72	32.97					
Ca ₈ Sr ₂ (PO ₄) ₆ (OH) ₂	$(2\ 1\ 1)$	13376	2.84	31.51	0.13	9.486(1)	6.959(2)	629 ± 15	87
	(0 0 2)	3885	3.48	25.57					
	(300)	7258	2.74	32.67		\ /			
Ca ₆ Sr ₄ (PO ₄) ₆ (OH) ₂	$(2\ 1\ 1)$	12594	2.87	31.19	0.23	9.546(4)	7.029(3)	354 ± 10	84
	(0 0 2)	3410	3.53	25.23		$\langle 1 \rangle$			
	(300)	6454	2.77	32.37					

 $d = \frac{0.9\lambda}{w\cos\theta}$

Crystallite size Debye-Scherrer

- w = full width at half maximum value (FWHM)
- $\lambda = 1.5405 \text{ \AA}$
- θ = diffraction angle at the (002) hkl reflection

$$X_{\rm c} \approx 1 - \frac{V_{112/300}}{I_{300}}$$

Crystallinity degree, fraction of crystalline phase, Xc

 I_{300} = intensity of (300) hkl reflection $V_{112}/_{300}$ = intensity of the hollow between (112) hkl and (300) hkl reflections.

Rietveld Refinement

Effect of Strontium on HA structure

•Increase of lattice parameters

Sr²⁺ 118 pm Ca²⁺ 100 pm

•Decrease in the crystallinity degree, consistent with the overall peak intensity decrease reported and also with Raman results

•Decrease in the specific surface area



Effect of Strontium on the morphology

After bioactivity test

large aggregates (hundreds of µm in size) of particles

J Mater Sci: Mater Med DOI 10.1007/s10856-012-4767-3

Magnesium- and strontium-co-substituted hydroxyapatite: the effects of doped-ions on the structure and chemico-physical properties

Valentina Aina · Gigliola Lusvardi · Basil Annaz · Iain R. Gibson · Flora E. Imrie · Gianluca Malavasi · Ledi Menabue · Giuseppina Cerrato · Gianmario Martra V. Aina · G. Cerrato · G. Martra Department of Chemistry, University of Turin, Via P. Giuria 7, 10125 Turin, Italy

V. Aina · G. Cerrato · G. Martra Centre of Excellence NIS (Nanostructured Interfaces and Surfaces), INSTM (Italian National Consortium for Materials Science and Technology), UdR University of Torino, Turin, Italy

G. Lusvardi (⊠) · G. Malavasi · L. Menabue Department of Chemistry, University of Modena and Reggio Emilia, Via Campi 183, 41125 Modena, Italy e-mail: gigliola.lusvardi@unimore.it

B. Annaz · I. R. Gibson · F. E. Imrie School of Medical Sciences, Institute of Medical Sciences, University of Aberdeen, Foresterhill, Aberdeen AB25 2ZD, UK

Magnesium : fundamental element and prevents possible risk factors for osteoporosis in humans

Strontium : improved solubility, antiresorptive activity, osteoclast apoptosis and osteoblast stimulation

Several articles have already been published on the chemico-physical properties of apatites and substituted apatites, *only a few works were devoted* to careful investigation of Mg-HA and *no paper reports* the study of co-substitution of Sr and Mg in HA.

The substituted samples are synthesized by an aqueous precipitation method

Ca(OH) ₂	H ₃ PO ₄	0.010.		(X - Mg and/or Sr)
	2 4	$Sr(NO_3)_2$	MgCl ₂ ·6H ₂ O	$(n = \log and or bi)$
0.100	0.06	-	-	1.67
0.099	0.06	-	0.001	1.67
0.095	0.06	-	0.005	1.67
0.090	0.06	-	0.010	1.67
0.090	0.06	0.009	0.001	1.67
0.090	0.06	0.005	0.005	1.67
0.080	0.06	0.019	0.001	1.67
0.080	0.06	0.015	0.005	1.67
0.080	0.06	0.010	0.010	1.67
10. 0				
Mallahan	undul	mmmm	v	Ca ₈ MgSr(PO ₄) ₅ (OH) ₂
Mound	muli	ron	~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~	Ca _{8.0} Mg _{0.1} Sr _{1.9} (PO ₄) ₅ (OH) ₂
Alleria	hala			Ca _{6.0} Mg _{0.5} Sr _{1.5} (PO ₄) ₆ (OH) ₂
	0.100 0.099 0.095 0.090 0.090 0.090 0.080 0.080 0.080	0.100 0.06 0.099 0.06 0.095 0.06 0.090 0.06 0.090 0.06 0.090 0.06 0.080 0.06 0.080 0.06 0.080 0.06 0.080 0.06	0.100 0.06 - 0.099 0.06 - 0.095 0.06 - 0.090 0.06 - 0.090 0.06 $0.0090.090$ 0.06 $0.0050.080$ 0.06 $0.0190.080$ 0.06 $0.0150.080$ 0.06 0.010	0.100 $0.060.099$ 0.06 - $0.0010.095$ 0.06 - $0.0050.090$ 0.06 - $0.0100.090$ 0.06 0.009 $0.0010.090$ 0.06 0.005 $0.0050.080$ 0.06 0.019 $0.0010.080$ 0.06 0.015 $0.0050.080$ 0.06 0.010 0.010



Samples	Phases ^a	Intensity(counts)	d (Å) (I ₁₂₁)	Position (°2 θ) (I ₁₂₁)	Crystallinity degree of HA $Xc \ (\% \pm 4)$	HA/ β -TCP	$a = b (\mathbf{\mathring{A}})^{\mathbf{b}}$	c (Å) ^b
						1410		
Ca10(PO4)6(OH)2	HA	7,806	2.82	31.77	96	- /	9.422(3)	6.880(5)
Ca _{9.9} Mg _{0.1} (PO ₄) ₆ (OH) ₂	HA	6,555	2.82	31.77	94		9.421(2)	6.878(5)
Ca _{9.5} Mg _{0.5} (PO ₄) ₆ (OH) ₂	HA	5,181	2.83	31.63	93	0 74/0.26	9.437(3)	6.870(4)
	β -TCP	1,827	2.87	31.14		/	10.346(1)	37.162(3)
Ca ₉ Mg ₁ (PO ₄) ₆ (OH) ₂	HA	2,991	2.80	31.98	89	0.48/0.52	9.423(2)	6.877(3)
	β-ΤСΡ	3,279	2.84	31.52			10.360(2)	37.174(5)
Ca ₉ Mg _{0.1} Sr _{0.9} (PO ₄) ₆ (OH) ₂	HA	7,356	2.84	31.53	91	0.81/0.19	9.501(4)	6.840(5)
	β-ΤСΡ	1,682	2.86	30.85			10.377(3)	37.167(3)
Ca ₉ Mg _{0.5} Sr _{0.5} (PO ₄) ₆ (OH) ₂	HA	4,785	2.85	31.37	80	0.94/0.06	9.463(5)	6.911(5)
	β-ΤСΡ	281	2.88	31.01			10.440(1)	37.635(6)
Ca8Mg0.1Sr1.9(PO4)6(OH)2	HA	4,166	2.84	31.44	80	0.74/0.26	9.493(4)	6.924(4)
	β-ΤСΡ	1,427	2.92	30.63			10.495(2)	37.846(5)
Ca8Mg0.5Sr1.5(PO4)6(OH)2	HA	2,958	2.85	31.35	73	0.54/0.46	9.489(2)	6.905(2)
	β-TCP	2,495	2.89	30.94			10.463(1)	37,332(3)
Ca ₈ Mg ₁ Sr ₁ (PO ₄) ₆ (OH) ₂	HA	2,732	2.85	31.44	68	0.46/0.54	9.510(3)	6.889(4)
	β-ΤСΡ	3,176	2.88	31.03			10.409(2)	37.385(3)

^a The structure model of these phases has been used for the Rietveld refinement

^b Theoretical values for HA and β -TCP are, respectively: a = b = 9.424(4) (Å), c = 6.879(4) (Å) and a = b = 10.429(2) (Å), c = 37.380(3) (Å)

Effect of Magnesium on HA structure

•Reduction of peak intensity and crystallinity

•At increasing amounts (0.5, 1 mol), HA and β -TCP

•HA/β-TCP ratio: from ~3 (75/25 %) to ~1 (50/50 %)

•contraction of lattice parameters especially evident in the β -TCP

Rietveld Refinement

Mg²⁺ (72 pm)

Samples	Phases ^a	Intensity(counts)	d (Å) (I ₁₂₁)	Position (°2 θ) (I ₁₂₁)	Crystallinity degree of HA Xc ($\% \pm 4$)	e HA/ β-TCP ratio	$a = b (\mathring{A})^{b}$	c (Å) ^b	
Ca10(PO4)6(OH)2	HA	7,806	2.82	31.77	96	-/	9.422(3)	6.880(5)	Rietveld
Ca _{9.9} Mg _{0.1} (PO ₄) ₆ (OH) ₂	HA	6,555	2.82	31.77	94	- +	9.421(2)	6.878(5)	Refinement
Ca _{9.5} Mg _{0.5} (PO ₄) ₆ (OH) ₂	HA	5,181	2.83	31.63	93	0.74/0.26	9.437(3)	6.870(4)	
	β -TCP	1,827	2.87	31.14		/	10.346(1)	37.162(3)	
Ca ₉ Mg ₁ (PO ₄) ₆ (OH) ₂	HA	2,991	2.80	31.98	89	0.48/0.52	9.423(2)	6.877(3)	
	β-ΤСΡ	3,279	2.84	31.52		1	10.360(2)	37.174(5)	
Ca ₉ Mg _{0.1} Sr _{0.9} (PO ₄) ₆ (OH) ₂	HA	7,356	2.84	31.53	91	0.81/0.19	9.501(4)	6.840(5)	
	β-ΤСΡ	1,682	2.86	30.85			10.377(3)	37.167(3)	
Ca9Mg0.5Sr0.5(PO4)6(OH)2	HA	4,785	2.85	31.37	80	0.94/0.06	9.463(5)	6.911(5)	
	β-ΤСΡ	281	2.88	31.01			10.440(1)	37.635(6)	
Ca8Mg0.1Sr1.9(PO4)6(OH)2	HA	4,166	2.84	31.44	80	0.74/0.26	9.493(4)	6.924(4)	
	β-TCP	1,427	2.92	30.63			10.495(2)	37.846(5)	
Ca8Mg0.5Sr1.5(PO4)6(OH)2	HA	2,958	2.85	31.35	73	0.54/0.46	9.489(2)	6.905(2)	
	β-TCP	2,495	2.89	30.94			10.463(1)	37.332(3)	
Ca8Mg1Sr1(PO4)6(OH)2	HA	2,732	2.85	31. <mark>4</mark> 4	68	0.46/0.54	9.510(3)	6,889(4)	
7	β-ΤСΡ	3,176	2.88	31.03			10.409(2)	37.385(3)	

^b Theoretical values for HA and β -TCP are, respectively: a = b = 9.424(4) (Å), c = 6.879(4) (Å) and a = b = 10.429(2) (Å), c = 37.380(3) (Å)

Effect of Magnesium and Strontium on HA structure

$Ca_{9}Mg_{0.1}Sr_{0.9}(PO_{4})_{6}(OH)_{2}$

Ca₉Mg_{0.5}Sr_{0.5}(PO₄)₆(OH)₂

•HA , $\beta\text{-TCP}$ •decreases the crystallinity degree of the HA phase •increases HA/ $\beta\text{-TCP}$ ratio

•Ca₉Mg_{0.1}Sr_{0.9}(PO₄)₆(OH)₂ -Lattice parameters: increase for the HA (effect of strontium), decrease for the β -TCP (effect of magnesium)





	Ca/P	Mg/P
e	xperimenta	l/theoretical
Ca10(PO4)6(OH)2		
ล	1.68/1.67	1
Ь	1.66/1.67	1
c	1.70/1.67	1
Ca9.9Mg0.1(PO4)6(OH)2		
a	1.66/1.65	0.02/0.02
ь	1.41/1.65	0.03/0.02
с	1.84/1.65	0.01/0.02
Ca25Mg05(PO4)6(OH)2		
a	1.62/1.58	0.09/0.08
b	1.35/1.58	0.11/0.08
c	1.77/1.58	0.04/0.08
Ca ₉ Mg ₁ (PO ₄) ₆ (OH) ₂		
a	1.49/1.50	0.15/0.17
b	1.26/1.50	0.18/0.17
c	1.60/1.50	0.07/0.17





the morphology is quite irregular in term of dimensions (5-40 lm) and shape (generally, rectangular and very sharp)



			A
<mark>, 50</mark> µп	Ca_M;	, Sr. (P	0 <u>)</u> (OII)

No.	Re	E	and a	
(then	AND I	30	hay!	
A.		100	X	
50 um	ia Mg.,	Sr _u (P	ojj (Oli),	
No and State	- 신장:	-		



	Ca/P	Mg/P	Sr/P
	experime	ental/theore	etical
Ca ₉ Mg _{0.1} Sr _{0.9} (PO ₄) ₆ (OH) ₂		0.0000000000000000000000000000000000000	
a	1.49/1.50	0.02/0.02	0.14/0.15
ь	1.25/1.50	0.02/0.02	0.17/0.15
c	1.56/1.50	0.03/0.02	0.02/0.15
Ca9Mg03Sr05(PO4)6(OH)2			
a	1.48/1.50	0.04/0.08	0.06/0.08
b	1.32/1.50	0.21/0.08	0.13/0.08
c	1.78/1.50	0.06/0.08	0.16/0.08
CasMg0.1Sr1.9(PO4)6(OH)2			
a	1.29/1.33	0.01/0.02	0.27/0.32
ь	1.21/1.33	0.02/0.02	0.24/0.32
c	1.60/1.33	0.01/0.02	0.30/0.32
CasMgosSr15(PO4)6(OH)2			
a	1.31/1.33	0.09/0.08	0.22/0.25
b	1.50/1.33	0.07/0.08	0.21/0.25
c	1.05/1.33	0.05/0.08	0.19/0.25
CasMg1Sr1(PO4)6(OH)2		0.000	
a	1.34/1.33	0.17/0.17	0.16/0.17
b	1.29/1.33	0.34/0.17	0.15/0.17
c	1.30/1.33	0.12/0.17	0.15/0.17

non homogeneous distribution of the elements that is consistent with the presence of two phases The combined use of XRPD, FTIR and Raman spectroscopies provides complementary data and indicates clearly phase formation and transformation as well as related structural changes resulting from the ions substituted in the HA structure

Magnesium and strontium, interact with both HA and β -TCP in terms of variation of lattice parameters and the degree of crystallinity of HA

A careful evaluation of the different molar ratios of the ions allows to obtain biphasic materials (BCPs) with selected ratios of HA and β -TCP, with also a controlled crystallinity degrees of HA

BCPs are favoured for clinical applications because their resorption rate can be tuned to match the bone healing rate allowing to obtain a suitable balance between implant degradation and bone regeneration

Key Engineering Materials Vols. 529-530 (2013) pp 88-93 © (2013) Trans Tech Publications, Switzerland doi:10.4028/www.scientific.net/KEM.529-530.88

Synthesis and Characterisation of Strontium and Magnesium Co-Substituted Biphasic Calcium Phosphates

Flora E. Imrie^{1,a}, Valentina Aina^{2,b}, Gigliola Lusvardi^{3,c}, Gianluca Malavasi^{3,d}, Iain R. Gibson^{1,e}, Giuseppina Cerrato^{2,f} and Basil Annaz^{1,g}

¹School of Medical Sciences, Institute of Medical Sciences, University of Aberdeen, Foresterhill, Aberdeen AB25 2ZD, United Kingdom

²Department of Chemistry, Università degli Studi di Torino, Via P. Giuria 7, 10125 Torino, Italy; INSTM (Italian National Consortium for Materials Science and Technology), UdR Università di Torino

³Department of Chemistry, University of Modena e Reggio Emilia, Via Campi 183, 41125 Modena, Italy

Information in a powder diffraction pattern



Rietveld Method

•Based on Whole Powder Profile Fitting (WPPF): all categories of observables (d_{hkl}, I_{hkl}, background.....) are considered together to build a model which is used to generate a full powder diffraction pattern



•Calculated pattern is compared to the observed pattern and modified by least squares in order to minimize the differences between observed and calculated

•Function minimized by least square is

 $\mathbf{S}_{\mathbf{y}} = \sum_{\mathbf{i}} \mathbf{w}_{\mathbf{i}} (\mathbf{y}_{\mathbf{oi}} - \mathbf{y}_{\mathbf{ci}})^2$

w_i = 1/y_i y_i = observed intensity y_{ci} = calculated intensity

•Least squares refinements are carried our until the best fit is obtained between the entire observed powder diffraction pattern taken as a whole and the entire calculated pattern

$y_{ci} = s \sum L_k |F_k|^2 \phi (2\theta_i - \theta_k) P_k A + y_{bi}$

- y_{ci} calculated intensity at the ith step
- s scale factor
- k the Miller indices, h, k, l, for a Bragg reflection
- $L_k \quad \mbox{multiplicity}, \mbox{Lorentz} \mbox{ and polarization factor }$
- F_k structure factor for the kth Bragg reflection
- $\phi \quad \ \ reflection \ profile \ function$
- P_k Preferred orientation
- A absorption factor
- y_{bi} background intensity at the ith step

Rietveld refinement fits the whole pattern and refines

- atomic positions
- lattice parameters
- profile parameters
- background parameters
- •instrumental parameters

Reasons to perform a Rietveld refinement

solving an unknown crystal structure

•quantitative determination of the percentages of different phases and quantification of amorphous content (QPA)

Rietveld Method in Practice

Basic requirements

- 1. Accurate diffraction data
- 2. A reasonable starting structural model
 - a. Space group symmetry
 - b. Approximate atomic positions
 - c. model may be from: isostructural materials, theoretical simulations, high-resolution atomic imaging
- 3. A Rietveld refinement program
 - a. GSAS (Larson and von Dreele)
 - b. Fullprof (Rodriguez Carvajal)
 - Others: BGMN (Bergmann), DBW (Wiles and Young), LHPM-Rietica (Hunter), MAUD (Lutterotti), Rietan (Izumi) Simref (Ritter)



Acknowledgments

```
University of Modena and Reggio Emilia
<u>Research group</u>
L.Menabue, M.C.Menziani , G.Malavasi, A.Pedone
```

```
University of Torino
V.Aina, L.Bergandi, V.Bolis, G.Cerrato, D.Ghigo, G.Martra
```

Universidad Complutense de Madrid M.V.Regì, A.J.Salinas

University of Aberdeen I.R.Gibson

